

Dipole-active optical phonons in YTiO₃: ellipsometry study and lattice-dynamics calculations

N. N. Kovaleva,^{1,2} A. V. Boris,^{1,2} L. Capogna,^{3,4} J. L. Gavartin,⁵

P. Popovich,¹ P. Yordanov,¹ A. Maljuk,⁶ A. M. Stoneham,⁷ and B. Keimer¹

¹*Max-Planck-Institut für Festkörperforschung, Heisenbergstrasse 1, D-70569 Stuttgart, Germany*

²*Department of Physics, Loughborough University, Loughborough, LE11 3TU, UK*

³*CNR-INFM, CRS-SOFT and OGG Grenoble, 6 Rue J. Horowitz, BP 156 F-38042 Grenoble CEDEX 9, France*

⁴*Institut Laue Langevin, 6 rue J. Horowitz BP 156 38042 Cedex 9 Grenoble, France*

⁵*Accelrys, Cambridge, CB4 0WN, UK*

⁶*Hahn-Meitner-Institute, Glienicker Str. 100, D-14109 Berlin, Germany*

⁷*London Center for Nanotechnology and Department of Physics and Astronomy, University College London, Gower Street, London WC1E 6BT, UK*

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The anisotropic complex dielectric response was accurately extracted from spectroscopic ellipsometry measurements at phonon frequencies for the three principal crystallographic directions of an orthorhombic (*Pbnm*) YTiO₃ single crystal. We identify all twenty five infrared-active phonon modes allowed by symmetry, 7B_{1u}, 9B_{2u}, and 9B_{3u}, polarized along the *c*-, *b*-, and *a*-axis, respectively. From a classical dispersion analysis of the complex dielectric functions $\tilde{\epsilon}(\omega)$ and their inverses $-1/\tilde{\epsilon}(\omega)$ we define the resonant frequencies, widths, and oscillator strengths of the transverse (TO) and longitudinal (LO) phonon modes. We calculate eigenfrequencies and eigenvectors of B_{1u}, B_{2u}, and B_{3u} normal modes and suggest assignments of the TO phonon modes observed in our ellipsometry spectra by comparing their frequencies and oscillator strengths with those resulting from the present lattice-dynamics study. Based on these assignments, we estimate dynamical effective charges of the atoms in the YTiO₃ lattice. We find that, in general, the dynamical effective charges in YTiO₃ lattice are typical for a family of perovskite oxides. By contrast to a ferroelectric BaTiO₃, the dynamical effective charge of oxygen related to a displacement along the *c*-axis does not show the anomalously large value. At the same time, the dynamical effective charges of Y and *ab*-plane oxygen exhibit anisotropy, indicating strong hybridization along the *a*-axis.

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I. INTRODUCTION

The richly diverse behavior of rare-earth titanates RTiO₃ with a single 3*d* electron in the nearly degenerate Ti *t*_{2*g*} orbitals is a consequence of the interplay between electron correlations, spin-orbit coupling, and electron-lattice coupling including the Jahn Teller (JT) effect [1, 2, 3, 4, 5, 6, 7, 8, 9, 10, 11, 12, 13, 14]. Fine adjustment of the perovskite lattice in the titanates, in accordance with the principle of close packing, occurs mainly due to the cooperative rotations of the TiO₆ octahedra, which are strongly dependent on the ionic radius of the rare-earth atom. The orbital structure that arises due to these distortions is determined by vibronic interactions between the lattice and the orbital subsystem. These octahedral distortions are also coupled to the magnetic ground state in RTiO₃: upon decreasing the ionic radius of *R* from La to Y, the transition temperature of the *G*-type antiferromagnetic state observed in LaTiO₃ (*T*_N = 150 K) shifts to progressively lower temperatures, and the ground state ultimately becomes ferromagnetic with *T*_C = 30 K in YTiO₃ [11, 15, 16, 17]. Point-charge [1, 9, 15] and density functional [12, 13] calculations reproduce many aspects of the physical properties of the titanates, including the magnetic ground states of the end-point compounds LaTiO₃ and YTiO₃, based on the

experimentally determined lattice structure. These calculations imply that the *t*_{2*g*} levels are split by ~ 100-200 meV, so that the Ti orbitals are almost frozen in the experimentally relevant temperature range. However, some aspects of the magnetic [4, 7] and orbital [18] excitation spectra are difficult to reproduce in the framework of these rigid-orbital models. This has stimulated theories based on substantial fluctuations in the orbital sector [5, 6]. While such fluctuations are not apparent in the specific heat [19], recent thermal conductivity measurements of RTiO₃ have been interpreted as evidence of orbital fluctuations [20]. Other recent studies of YTiO₃ single crystals have revealed anomalous temperature dependencies of the optical spectral weight [21] and thermal expansion [22] with an onset around 100 K, well above *T*_C, which may be manifestations of a temperature-dependent rearrangement of the *t*_{2*g*} orbitals.

Because of the intimate relationship between lattice structure, orbital occupation, and magnetism, the lattice dynamics plays an important role in the unusual properties of the rare-earth titanates. In particular, one would expect anharmonic behavior and damping of the most strongly coupled modes in the spin-lattice-orbital sector due to orbital fluctuations, as well as anomalous energy shifts of phonons around the magnetic transition temperature [23, 24]. Therefore experimental and theoretical studies of the lattice dynamics have the potential to be-

come quantitative diagnostics of the mechanisms underlying the unusual phenomena in the rare-earth titanates. However, as little information about the phonon spectra and the assignment of the phonon modes of $RTiO_3$ is thus far available, this potential has not yet been exploited.

The approach described here combines theoretical lattice-dynamics calculations with accurate spectroscopic ellipsometry measurements on well-characterized, nearly stoichiometric $YTiO_3$ single crystals of high quality. From the ellipsometry experiments we obtain the frequencies and oscillator strengths of all infrared (IR)-active TO phonon modes allowed by the $Pbnm$ symmetry. The lattice-dynamics calculations have been performed using the GULP code [25]. We calculate the eigenfrequencies of the Brillouin-zone center normal modes of B_{1u} , B_{2u} , and B_{3u} symmetry and the corresponding eigenvector components. Based on these calculations, we suggest assignments of the TO phonon modes observed in our ellipsometry spectra by comparing their frequencies and oscillator strengths with the calculated values. In addition, we estimate the dynamical effective charges of the atoms in the $YTiO_3$ lattice using the calculated eigenvector components and the experimental phonon parameters of the TO frequencies and oscillator strengths. The assignment is further verified by *ab initio* lattice-dynamics calculations [26] within the density functional approach using the VASP software package [27, 28]. The present study provides a coherent picture of the lattice vibrations in $YTiO_3$, consistent with both experimental data and with full electronic structure calculations.

II. CRYSTAL STRUCTURE AND Γ -POINT PHONONS IN $YTiO_3$

$YTiO_3$ exhibits an orthorhombic structure of a $GdFeO_3$ type described by the $Pbnm$ space group ($Pnma D_{2h}^{16}$, No. 62 in the International Tables for X-ray Crystallography [29], with nonconventional coordinate axes orientation, $Pnma(a, b, c) \iff Pbnm(b, c, a)$), with 4 f.u./unit cell (see Fig. 1, on the right side). This orthorhombic structure can be regarded as a distorted modification of the simple cubic ABO_3 perovskite structure with space group $Pm\bar{3}m$ (see Fig. 1, on the left side). The orthorhombic distortions can be attributed to the common atomic size mismatch: the sum of the ionic radii in the Ti-O₂ layer, $r_{Ti} + r_O$, does not match that of the Y-O layer, $(r_Y + r_O)/\sqrt{2}$, in the right way for a stable cubic structure. These orthorhombic distortions lead first to the oxygen ion displacements, resulting from the two consequent and coordinated rotations of the TiO_6 octahedra, described by $a^-a^-b^+$ in the Glazer's notations [30], and second to a modification of the Y-O and Y-Ti coordinations. In addition, there are JT-type distortions, resulting in a slight extension or compression of the Ti-O bond pairs, and the “scissors”-type distortions of the TiO_6 octahedron [11, 14, 15]. The Wyckoff

positions of the atoms and their site symmetries in the orthorhombic unit cell are listed in Table 1. The irreducible representations (in the $Pbnm$ notation) corresponding to various atomic sites in the orthorhombic structure that follow from the character tables of the point groups are presented in the right column of Table 1. The total numbers of the modes are grouped according to their optical activity. Among the total number of 60 Γ -point phonons, 24 ($7A_g + 7B_{1g} + 5B_{2g} + 5B_{3g}$) are Raman-active modes, 25 ($7B_{1u} + 9B_{2u} + 9B_{3u}$) are IR-active modes, 8 (A_u) are silent modes, and 3 ($B_{1u} + B_{2u} + B_{3u}$) are acoustic modes.

III. SAMPLES AND EXPERIMENTAL APPROACH

A. Crystal growth and characterization

It has to be emphasized that $YTiO_{3+\delta}$ single crystals are always slightly off-stoichiometric, having an excess of oxygen above the exact stoichiometry formula. As to our knowledge, a minimal level record of $\delta = 0.009$ belongs to single crystals of $YTiO_{3+\delta}$ grown by Czochralski method [31]. In this study, single crystals of $YTiO_3$ were grown by the floating zone method in a reducing atmosphere ($Ar/H_2 = 50/50$). More details are given in Ref. [21]. The oxygen excess in our samples was measured by the Differential Thermal Analysis and Thermal Gravimetry Analysis and estimated at a level less than $\delta = 0.013$. The same level of the oxygen off-stoichiometry $\delta = 0.01$ was determined in the single crystals of $YTiO_{3+\delta}$, grown by the floating zone method in a fairly strong reducing condition ($Ar/H_2 = 70/30$) by Okimoto *et al.* [32]. The optical conductivity spectra measured on our crystals and those grown by Okimoto *et al.* are very well agree with each other in a wide spectral range, from the phonon frequencies up to the deep UV frequencies (see Fig. 7 in Ref. [21] and Fig. 2 in Ref. [32]). Therefore, we conclude that our single crystals of $YTiO_3$ grown by the floating zone method are of the same quality, verified by the optical probe. Moreover, the Ti^{3+} ions in the $YTiO_3$ lattice are unstable in oxygen atmosphere, and fully oxidize to Ti^{4+} above 300 °C in flowing oxygen (see Fig. 2 in Ref. [21]). This may lead to the aging effects in air, associated with the formation of the titanium oxides with the valence of Ti increased above 3+ and simultaneous incorporation of extra oxygen (interstitials). The aging effects and the oxygen off-stoichiometry in $YTiO_{3+\delta}$ crystals have a deep impact on the optical conductivity spectra and their temperature dependencies [21, 33]. If the samples are kept in air for extending periods of time, the weak satellites can appear in the phonon modes, as arising from the coupling with the local oxygen modes associated with oxygen interstitials, as in the case of the *ab* surface, reported in Ref. [21]. Therefore, an accurate study of the optical properties in $YTiO_{3+\delta}$ crystals should be always referred to the oxygen off-stoichiometry δ .

The samples were further characterized by magnetometry using a superconducting quantum interference device. For the almost stoichiometric $\text{YTiO}_{3+0.013}$ sample studied here, we estimate a Curie temperature of $T_C = 30$ K, an inflection point in the temperature-dependent magnetization is observed at 27 K. The determined saturated magnetic moment in the easy direction at 7 T is $\mu^c = 0.84 \mu_B$, whereas $\mu^b = 0.82 \mu_B$ in the hard direction [21]. These our data are in good agreement with the magnetization data on the nearly oxygen stoichiometric YTiO_3 single crystals, which referred to T_C of 29 K [34] and 30 K [35, 36].

At the same time x-ray diffraction measurements show that the YTiO_3 single crystals used for our experiments are untwinned, and posses an excellent crystal quality with mosaicity less than 0.03° . At room temperature, the lattice parameters are $a = 5.331(3)$, $b = 5.672(4)$, and $c = 7.602(6)$ Å, slightly different from those of Czochralski-grown single crystals reported by Maclean *et al.* [31]. The samples were aligned along the principal axes and cut in the form of a parallelepiped with dimensions $\sim 3 \times 3 \times 3$ mm³.

B. Ellipsometry approach

For optical measurements the sample surfaces were polished to optical grade. The technique of ellipsometry provides significant advantages over conventional reflection methods in that (i) it is self-normalizing and does not require reference measurements and (ii) $\epsilon_1(\omega)$ and $\epsilon_2(\omega)$ are obtained directly without a Kramers-Kronig transformation [37, 38, 39]. For ellipsometry measurements at phonon frequencies 50-700 cm⁻¹ (0.006-0.087 eV) we used the home-built ellipsometer in combination with a fast-Fourier transform interferometer (Bruker IFS 66v/S) at the infrared beam line of the ANKA synchrotron at the Forschungszentrum Karlsruhe, Germany. The high brilliance of the synchrotron light source due to the small beam divergence enabled us to perform very accurate ellipsometric measurements in the far-IR range with incidence angles ranging from 65° to 85° . For low-temperature measurements the sample was mounted on the cold finger of a helium flow cryostat. With only a single angle of incidence, the raw experimental data are represented by real values of the ellipsometric angles, Ψ and Δ , for any wave number. These values are defined through the complex Fresnel reflection coefficients for light polarized parallel (r_p) and perpendicular (r_s) to the plane of incidence

$$\tan \Psi e^{i\Delta} = \frac{r_p}{r_s}. \quad (1)$$

To determine the complex dielectric response $\tilde{\epsilon}_a(\omega)$, $\tilde{\epsilon}_b(\omega)$, and $\tilde{\epsilon}_c(\omega)$ of the YTiO_3 crystal, we measured four high-symmetry orientations of the ac and bc surfaces, with a or b axes aligned either parallel or perpendicular to the plane of incidence of the light, respectively.

Note here that we have used an overdetermined data set to check the consistency of our results. A nonlinear fitting procedure has been applied to extract point-by-point the complex dielectric response throughout the covered spectral range [40]. In the following, we present the complex dielectric response $\tilde{\epsilon}(\omega)$ extracted from the raw ellipsometry spectra, $\Psi(\omega)$ and $\Delta(\omega)$, according to this approach.

IV. RESULTS AND DISCUSSION

A. Infrared spectra of YTiO_3

Figures 2-4 show the dielectric functions $\tilde{\epsilon}(\omega)$ and their inverses $-1/\tilde{\epsilon}(\omega)$ for the three principal crystallographic axes of the YTiO_3 crystal, measured with an angle of incidence of 75° at $T = 25$ K. As discussed in Sec. II, a factor-group analysis for the orthorhombic crystal structure of YTiO_3 (space group $Pbnm$, D_{2h}^{16}) with 4 f.u./unit cell yields a total number of 25 ($7B_{1u}+9B_{2u}+9B_{3u}$) IR-active modes. Our polarized ellipsometry measurements allowed us to observe all symmetry-allowed IR-active phonon modes. One can easily identify six phonon modes in the polarization along the c axis (vibrations of B_{1u} symmetry), eight phonon modes in the b axis (B_{2u}), and eight phonon modes in the a axis (B_{3u}). In addition, we were able to identify very weak features in our raw ellipsometry spectra, $\Psi(\omega)$ and $\Delta(\omega)$, on the high-frequency side in each polarization (not shown), which we associate with $B_{1u}(7)$, $B_{2u}(9)$, and $B_{3u}(9)$ symmetry modes. Using the classical dispersion analysis we fit the complex dielectric functions $\tilde{\epsilon}_a(\omega)$, $\tilde{\epsilon}_b(\omega)$, and $\tilde{\epsilon}_c(\omega)$ (and their inverses independently) with a set of Lorentzian oscillators

$$\tilde{\epsilon}(\omega) = \epsilon_\infty + \sum_j \frac{S_j \omega_j^2}{\omega_j^2 - \omega^2 - i\omega\gamma_j}, \quad (2)$$

where ω_j , γ_j , and S_j are the resonant frequency, width, and dimensionless oscillator strength of the j th oscillator, and ϵ_∞ is the core contribution from the dielectric function [41]. The resulting parameters determine $\omega_{\text{TO}j}$, $\gamma_{\text{TO}j}$, and S_j of the transverse TO phonon modes (and $\omega_{\text{LO}j}$ and $\gamma_{\text{LO}j}$ of the longitudinal LO phonon modes). We determine the anisotropic static dielectric constants ϵ_0 from our fit at 50 cm⁻¹ (these values are given in Table IV), and estimate the anisotropic high-frequency dielectric constants defined by the relation $\epsilon_0 = \epsilon_\infty^* \prod_m \frac{\omega_{\text{LO},m}^2}{\omega_{\text{TO},m}^2}$. In Table II we list all Lorentzian parameters, together with ϵ_∞ and ϵ_∞^* , which show slightly different values, according to the accuracy of the present study.

B. Theoretical calculations

1. Shell model approximation and potential parameters

To determine the eigenfrequencies and eigenvectors of the normal modes and to make an assignment in the experimental phonon spectra, we have performed lattice-dynamics calculations for the orthorhombic YTiO_3 structure in the $Pbnm$ (D_{2h}^{16}) space group in the framework of a shell model. The shell model approach has been successfully applied earlier to study various physical properties and lattice dynamics of transition metal oxides, including high- T_c superconductors [42, 43, 44] and LaMnO_3 and YMnO_3 [45, 46, 47]. Our calculations were performed in the GULP code [25], using the lattice constants and fractional coordinates of the atoms reported for the Czochralski-grown single crystals by Maclean *et al.* [31]. In the shell model approximation, the lattice is considered as an assembly of polarizable ions, represented by massive point cores and massless shells coupled by isotropic harmonic forces defined by the spring constants. The interaction includes contributions from Coulomb and short-range interactions. The short-range potentials used for the shell-shell interactions are of the Buckingham form

$$V_{ij} = A_{ij} \exp(-r/\rho_{ij}) - C_{ij}/r^6. \quad (3)$$

The O-O (shell-shell) interactions are taken as those of typical oxides [48, 49]. We adopt the formal ionic charges here. The forces acting on the atoms in the lattice were minimized by varying the Buckingham potential parameters, spring constants, as well as the core and shell charges. The crystalline structure was allowed to relax to equilibrium conditions under the symmetry ($Pbnm$) restrictions, the cores and shells were allowed to move separately. The resulting Buckingham potential parameters, shell charges (Y) and force constants (k) are presented in Table III. The criterion for a successful fit was good agreement with the structural parameters [31], provided that approximately zero forces are applied for all atoms, and with the experimental anisotropic static ϵ_0 and high-frequency ϵ_∞ dielectric constants, determined in the present spectroscopic ellipsometry study (as Table IV illustrates).

2. Calculated Γ -point IR-active modes in YTiO_3

A factor group analysis of the zone-center phonon modes in the orthorhombic YTiO_3 ($Pbnm$) structure yields 25 IR-active phonon modes, which have their progenitors in the Brillouin-zone of the cubic perovskite ABO_3 ($Pm\bar{3}m$) structure [50]. Among them, there are nine phonon modes, $3B_{1u} + 3B_{2u} + 3B_{3u}$, emanating from the three IR-active F_{1u} cubic triplets. These are commonly referred to the *external* mode (in which the BO_6 octahedron vibrates against the A atoms), the

bending mode (B-O-B bond-angle modulation), and the *stretching* mode (B-O bond-length modulation). In addition, two IR-active modes of B_{2u} and B_{3u} symmetry and one silent mode of A_u symmetry stem from the cubic F_{2u} silent triplet. The rest orthorhombic IR modes arise from X , M , and R points of the cubic Brillouin-zone, and become active due to Brillouin-zone folding in the orthorhombic phase [50]. However, in this study we are not aimed at a detailed description of the orthorhombic phonon eigenvectors in terms of the corresponding cubic eigenvectors. Here we present results of our lattice-dynamics shell-model (SM) calculations and make an assignment of the IR-active phonon modes $\omega_{\text{TO}j}$ observed in the polarized ellipsometry spectra $\tilde{\epsilon}(\omega)$ in the YTiO_3 single crystal (see Figs. 2-4 and Table II).

We tentatively divide the observed phonon modes into three groups centered at about the characteristic frequencies of *external* (170 cm^{-1}), *bending* (340 cm^{-1}), and *stretching* modes (540 cm^{-1}) of the rare-earth perovskite oxides. Correspondingly, the eigenvector patterns for all normal modes of each symmetry B_{1u} , B_{2u} , and B_{3u} , as derived from the SM lattice-dynamics calculations, are presented in Figs. 5-7. Here only major components of the eigenvectors (the whole list is given in Table V) along the principal crystallographic directions are displayed for 20 atoms in the unit cell, of different symmetries and in all inequivalent positions. At each pattern the calculated eigenfrequencies are indicated. As one can notice from Table VI, the calculated eigenfrequencies generally well agree with the experimental TO phonon frequencies $\omega_{\text{TO}j}$. The character of the B_{1u} , B_{2u} , and B_{3u} modes is determined according to the dominant contribution from the collective vibrations of Y, Ti, O1, or O2 atoms, polarized along the c -, b -, and a -axis, respectively. However, mixing due to out-of-phase vibrations of ions of equal symmetry in inequivalent positions often does not allow the determination of a unique character for a given set of equal symmetry modes.

At low frequencies, near the characteristic frequency of the cubic perovskite *external* mode, one can identify three modes, $B_{1u}(1) : 167 \text{ cm}^{-1}$, $B_{2u}(2) : 240 \text{ cm}^{-1}$, and $B_{3u}(2) : 209 \text{ cm}^{-1}$, which exhibit very high oscillator strengths in the experimental IR phonon spectra, in agreement with the SM calculations (see Table VI). The eigenvectors of the $B_{1u}(1)$ mode show that the oscillator strength of this mode is determined by vibrations of the TiO_6 octahedra against the Y atoms along the c -axis. The eigenvectors of the $B_{2u}(2)$ and $B_{3u}(2)$ modes show that these normal modes involve dominant contribution from the collective Y-Ti vibrations, where Y and Ti atoms move in the opposite directions, polarized in accordance with the mode symmetries along the b - and a -axis, respectively. The eigenvectors of the $B_{2u}(1) : 139 \text{ cm}^{-1}$ and $B_{3u}(1) : 154 \text{ cm}^{-1}$ modes indicate that these modes are primarily due to out-of-phase vibrations of different Y atoms and out-of-phase vibrations of different Ti atoms, polarized respectively along the a - and b -axis, not coincident with the mode symmetries. In agreement with

the SM calculations, the observed phonon modes $B_{2u}(1)$ and $B_{3u}(1)$ exhibit very small oscillator strengths. Another weak mode is the c -axis $B_{1u}(2)$: 204 cm^{-1} , which is mainly due to b -axis in-phase vibrations of Ti atoms in one plane and out-of-phase vibrations in different planes.

At medium phonon frequencies near the cubic perovskite *bending* mode, the strong c -axis polarized modes $B_{1u}(3)$: 324 cm^{-1} and $B_{1u}(5)$: 387 cm^{-1} are mainly due to Ti-O2-Ti *bending* vibrations (the $B_{1u}(3)$ mode has small admixture of O1 character due to the movement of O1 atoms in-phase with Ti atoms). The b -axis polarized modes $B_{2u}(3)$: 273 cm^{-1} and $B_{2u}(7)$: 461 cm^{-1} involve collective vibrations of O2 and Y atoms moving in the opposite directions. In turn, the strong a -axis polarized modes $B_{3u}(3)$: 316 cm^{-1} and $B_{3u}(6)$: 426 cm^{-1} also involve collective vibrations of O2 atoms (the $B_{3u}(3)$ mode has in addition small contribution due to in-phase vibrations of O1 atoms and out-of-phase vibrations of Y atoms). The character of the mode $B_{2u}(4)$: 308 cm^{-1} is defined by the dominant contribution from O1 vibrations, moving out-of-phase with O2 atoms. The modes $B_{2u}(5)$: 364 cm^{-1} and $B_{3u}(5)$: 344 cm^{-1} can be assigned as Ti-O1-Ti *bending* modes. Other modes in this range $B_{2u}(6)$: 380 cm^{-1} and $B_{3u}(7)$: 516 cm^{-1} are also determined by O1 collective vibrations, polarized according to the mode symmetries. We note here that the two b -axis modes $B_{2u}(5)$: 364 cm^{-1} and $B_{2u}(6)$: 380 cm^{-1} having very close frequencies could be reversed in the assignment, as it is explained below. The c -axis $B_{1u}(4)$: 349 cm^{-1} has very small oscillator strength as it can be assigned to the a -axis out-of-phase vibrations of Ti atoms.

In the third group, the strong modes $B_{1u}(6)$: 545 cm^{-1} , $B_{2u}(8)$: 528 cm^{-1} , and $B_{3u}(8)$: 554 cm^{-1} are observed in the experimental IR phonon spectra. Therefore, these IR-active phonon modes of orthorhombic YTiO_3 should be most closely related to the cubic perovskite *stretching* modes. Indeed, the eigenvector patterns show that the $B_{1u}(6)$ mode is due to the c -axis Ti-O1 vibrations resulting in the Ti-O1 bond-length modulation, and the oscillator strengths of the $B_{2u}(8)$ and $B_{3u}(8)$ normal modes are determined by the b - and a -axis Ti-O2 collective out-of-phase vibrations. In addition, we were able to identify very weak features in our ellipsometry spectra on the high-frequency side in each polarization, which we associate with the $B_{1u}(7)$: 595 cm^{-1} , $B_{2u}(9)$: 577 cm^{-1} , and $B_{3u}(9)$: 576 cm^{-1} symmetry modes. We note that for the $B_{1u}(6)$ and $B_{1u}(7)$ modes there is a good agreement between the frequencies and oscillator strengths, predicted from the calculations and those observed (see Table VI). The $B_{1u}(7)$ normal mode is weak, because it is mostly due to the different symmetric *ab*-polarized out-of-phase oscillations of Ti and O2 atoms. However, our SM calculations predict that the strong in-plane modes of B_{2u} and B_{3u} symmetry appear at the high-frequency boundary of the phonon spectra. In this case, in our assignment we relate the modes of similar oscillator strengths. In agreement with our SM calculations

the $B_{2u}(9)$ and $B_{3u}(9)$ normal modes are largely due to the symmetric a - and b -polarized out-of-phase vibrations of Ti and O1 atoms, respectively. Regarding the symmetry of these modes, their oscillator strengths is expected to be small, in consistency with our experimental observations.

3. Comparison with first-principles results

In order to independently support the assignment of the IR-active phonon modes in YTiO_3 , we have calculated the Γ -point normal modes by means of the *ab initio* approach within the density functional theory using the VASP software package [27, 28]. A detailed description of these calculations will be given elsewhere [26]. Here we only mention the eigenfrequencies of the normal modes resulting from the *ab initio* calculations: a -axis 9 B_{3u} normal modes are calculated at the frequencies 137, 175, 298, 312, 330, 379, 391, 473, and 519 cm^{-1} ; b -axis 9 B_{2u} normal modes at 125, 211, 250, 291, 337, 359, 440, 485, and 500 cm^{-1} ; and c -axis 7 B_{1u} normal modes at 144, 182, 289, 336, 352, 463, and 517 cm^{-1} . In general, the frequencies and eigenvectors of the normal modes predicted by the *ab initio* lattice-dynamics calculations are in a reasonable agreement with the results of the shell model lattice-dynamics calculations reported in detail here, especially for the low-frequency and high-frequency modes, near the characteristic frequencies of the cubic perovskite *external* and *stretching* modes, respectively. The main characters of the modes near the characteristic cubic perovskite *bending* mode (where the phonon frequencies are densely spaced and therefore the results are very sensitive to details) are also in satisfactory agreement with the shell model lattice-dynamics calculations. For example, the assignments of the b -axis modes $B_{2u}(3)$: 273 cm^{-1} and $B_{2u}(4)$: 308 cm^{-1} , the a -axis mode $B_{3u}(6)$: 426 cm^{-1} and the c -axis mode $B_{1u}(3)$: 324 cm^{-1} are well reproduced.

4. Effective charges in YTiO_3

The *static* effective charges $Z_i e$ related to the bonding ionicity can be defined for a ternary compound as follows [51, 52]

$$\sum_j [\omega_{\text{LO}j}^2 - \omega_{\text{TO}j}^2] = \frac{4\pi}{V_c} \sum_i (Z_i e)^2 / M_i, \quad (4)$$

where j denotes the phonon mode with TO frequency $\omega_{\text{TO}j}$ and LO frequency $\omega_{\text{LO}j}$, V_c is the unit cell volume, M_i corresponds to the mass of the i th atom. The static effective charges of oxygen $Z_{\text{O}e}$ can be estimated rather accurately from this relation, as the contribution of the terms related to oxygen is dominant on the right hand side of Eq. (4) due to the small mass of oxygen as compared with the masses of other constituents. Using the

frequencies $\omega_{\text{TO}j}$ and $\omega_{\text{LO}j}$ determined from the ellipsometry measurements listed in Table II, we have found that the static effective charges of oxygen in orthorhombic YTiO_3 are nearly isotropic: $Z_{O\parallel a} = -1.14 |e|$, $Z_{O\parallel b} = -1.17 |e|$, and $Z_{O\parallel c} = -1.20 |e|$. These values are typical for the entire family of perovskite oxides.

The Born effective charge of an ion represents the dynamical contribution to the effective charge. It is controlled by long-range Coulomb interactions. The *dynamical* effective charge is represented by a tensor, since it is defined by the change of macroscopic polarization created in direction β , induced by the periodic displacement $\tau_{\kappa,\alpha}$ of the sublattice of atoms κ in direction α at the linear order, times the unit cell volume

$$Q_{\kappa,\alpha\beta}^* = V_c \frac{\partial \mathcal{P}_\beta}{\partial \tau_{\kappa,\alpha}}. \quad (5)$$

For the case of ABO_3 compounds, the values of the Born effective charges deviate substantially from the static effective charges. The *ab initio* calculations report anomalously large values of dynamical effective charges for Ti and O ions in ferroelectric perovskites KNbO_3 and BaTiO_3 [53, 54]. The charges of Ba and Ti atoms in the orthorhombic structure Q_{Ba}^* and Q_{Ti}^* of BaTiO_3 are nearly isotropic and vary, respectively, in the range from +2.72 to +2.77 and from +5.59 to +6.8. In turn, for oxygen, the effective charges exhibit predominantly uniaxially anisotropic character, where the values of $Q_{O\parallel}^*$ vary from -1.91 to -2.04 and $Q_{O\perp}^*$ vary from -4.89 to -5.45, and refer, respectively, to a displacement of oxygen in the *ab*-plane and along the *c*-axis [54].

The oscillator strengths S_j of IR-active phonon modes are related to the dipole moment arising from the displacements of ions involved in a vibration

$$S_j = \frac{4\pi}{V_c} \frac{[\sum_i Q_{Ti}^* u_{ij}]^2}{\omega_{\text{TO}j}^2 \sum_i M_i u_{ij}^2}. \quad (6)$$

Here Q_{Ti}^* corresponds to the dynamical effective charge of the *i*th atom vibrating in *j* mode with the TO frequency $\omega_{\text{TO}j}$, and u_{ij} represents the displacements related to the eigenvector χ_j by $u_{ij} = \chi_{ij}/\sqrt{M_i}$. In Table VI we present calculated squares of net dipole moments $[\sum_i Q_i u_{ij}]^2$ arising from the displacements of all ions involved in the different symmetry IR-active vibrations, using the formal ionic charges that were adopted in our shell model. These values are juxtaposed with the corresponding values estimated from the experimental phonon parameters of $\omega_{\text{TO}j}$ and oscillator strengths S_j given in Table II. As one can see, the qualitative agreement between the model and the experiment is quite good in the sense that the model correctly predicts the strongest modes in the phonon spectra and the weak modes are also predicted to be weak. However, some of the modes, for example, two strong *b*-axis modes $B_{2u}(3) : 273 \text{ cm}^{-1}$ and $B_{2u}(6) : 380 \text{ cm}^{-1}$, two strong *a*-axis modes $B_{3u}(2) : 209 \text{ cm}^{-1}$, $B_{3u}(6) : 426 \text{ cm}^{-1}$, and one weak mode $B_{3u}(7) : 516 \text{ cm}^{-1}$, and the strong *c*-axis mode $B_{1u}(3)$

: 324 cm^{-1} show largest discrepancy between the experimental values and calculated in the framework of the adopted shell model.

An improved description of the experimental oscillator strengths of all phonon modes compared to those with formal ionic charges can be achieved by introducing Born effective charges. These effective charges can be extracted by inversion the linear equations Eq. (6) by making use of the eigenvectors, providing the experimental phonon parameters of $\omega_{\text{TO}j}$ and S_j are available. We note here that for each polarization together with the electric charge neutrality condition, we have an over determined data set for the four unknown effective charges. However, the inversion of Eq. (6) results in large errors if it is applied for the modes which have eigenvector components close to zero, in particular, for the weak modes (see Tables V and VI), as the determinant of the matrix is then close to zero. We apply Eq. (6) to the three strong modes in each polarization and determine the corresponding dynamical effective charges, which are listed in Table VII. In reasonable agreement with the effective charges of Ti atoms in the orthorhombic structure of BaTiO_3 , with the nominal valence of Ti atoms +4 [54], we have found that in YTiO_3 the dynamical effective charges of Q_{Ti}^* are nearly isotropic, varying in the range from +3.9 to +5.00. At the same time, the effective charges of Y and O2 atoms exhibit *a*-axis anisotropy: $Q_{Y\parallel a}^* = +1.5$ and $Q_{O2\parallel a}^* = -1.9$, compared to the corresponding effective charges $Q_{Y\parallel b,c}^*$ of $+3.4 \div +3.9$ and $Q_{O1,2\parallel b}^*$ of $-2.5 \div -2.7$. The dynamical effective charges of oxygen $Q_{O1,2\perp}^*$, related to a displacement along the *c*-axis, vary in the range from -2.3 to -2.8 in YTiO_3 and do not show anomalously large values like in ferroelectric BaTiO_3 [54]. In Table VI the calculated values of the $[\sum_i Q_i^* u_{ij}]^2$ are juxtaposed with the corresponding experimental values and those obtained from the calculations with formal ionic charges. One can notice that, in general, the agreement with the experimental values is improved (except for some of the weak modes, which are less accurately predicted). The description with the effective charges give also an indication that two *b*-axis modes $B_{2u}(5) : 364 \text{ cm}^{-1}$ and $B_{2u}(6) : 380 \text{ cm}^{-1}$ having very close frequencies should be reversed in the assignment. The agreement for the *a*-axis modes $B_{3u}(6) : 426 \text{ cm}^{-1}$ and $B_{3u}(7) : 516 \text{ cm}^{-1}$ is not substantially improved by introducing Born effective charges. A possible reason could be a strong dependence of the mode $B_{3u}(7)$ on strain in the real crystal structure. As a result, most of its oscillator strength might be transferred to the strongest *a*-axis mode $B_{3u}(6)$. We note that the *b*-axis mode $B_{2u}(3) : 273 \text{ cm}^{-1}$ is less successfully predicted, however, the character of this mode is well reproduced by the present *ab initio* calculations [26].

V. CONCLUSIONS

In summary, we present optical phonon spectra of the orthorhombic YTiO_3 single crystal as obtained by spec-

troscopic ellipsometry and suggest the assignments of all symmetry-allowed zone-center IR-active phonon modes by comparing them with the results of the shell model calculations. We note that the shell model elaborated in this work may possess certain advantages, as the parameters of the model were fitted not only to the orthorhombic crystal structure, but also to the experimental anisotropic static ϵ_0 and high-frequency ϵ_∞ dielectric constants, determined in the present ellipsometry study. Consequently, the calculated dipole moments (or oscillator strengths) associated with displacements of the atoms involved in the IR-active normal modes in YTiO₃ are in good agreement with the experimental data. The normal modes predicted in the framework of the present shell model are in good agreement with the *ab initio* lattice-dynamics calculations [26]. Thus, our study gives us a coherent picture of the lattice vibrations in YTiO₃, consistent with both experimental data and with full electronic structure calculations. We have not considered the influence of magnetic interactions on the lattice dynamics, but as the ferromagnetic coupling in YTiO₃ is weak it should not affect our results substantially.

In order to describe the lattice dynamics of complex oxides like the rare-earth titanates, one can easily imagine a need to go beyond the shell model combined with interatomic potentials. However, the shell model itself is a general model within the dipole and harmonic approximations [55], and there have been wide-ranging and successful calculations for simple binary oxides [56], including predictions of surface structure and vibrations, off-center ions, charge transfer transitions, and defect equilibria for non-stoichiometric oxides. What was unexpected is the extent to which the oxygen-oxygen potentials seem to be transferrable to the more complex oxides. One might have expected the Jahn-Teller effect or charge transfers and lack of ionicity to lead to radical changes. This seems not to be the case: we find the O-O potentials based on MgO and similar binary oxides essentially the same for LaMnO₃, for Bi-based oxide superconductors, and for YTiO₃. Some of the reasons for this are discussed in relation to oxide superconductors in Ref. [57]. As a result, the shell model combined with interatomic potentials continues to offer a powerful route to understanding experimental data for complex oxides.

The comprehensive study described here provides a solid foundation for further quantitative exploration of the interplay between spin, orbital and lattice degrees of freedom in the titanates. We found that in YTiO₃ the effective charges of Ti are nearly isotropic and comparable to the effective charges of Ti in other perovskite compounds. At the same time the effective charges of Y and O2 atoms exhibit *a*-axis anisotropy: $Q_{Y\parallel a}^* = +1.5$ and $Q_{O2\parallel a}^* = -1.9$, compared to the corresponding effective charges $Q_{Y\parallel b,c}^*$ ($+3.4 \div +3.9$) and $Q_{O1,2\parallel b}^*$ ($-2.5 \div -2.7$). As the effective charges are a sensitive tool for analyzing dynamic changes of orbital hybridization, we suggest strong hybridization effects between Y and *ab*-plane oxygen along the *a*-axis. Hybridization of Y 5*d* and O 2*p* states has indeed been predicted by density functional calculations [13]. Then, according to our assignment, the modes B_{3u}(3) : 316 cm⁻¹ and B_{3u}(6) : 426 cm⁻¹, involving O2 and Y displacements along the *a* axis, will probably exhibit anomalous temperature dependencies. Some of the Raman-active modes related to the “scissors”-type distortions of the TiO₆ octahedra, which are especially important in YTiO₃ [14], might also exhibit anomalous temperature dependence in the vicinity of T_C . As in the real structure of YTiO₃ these distortions are most pronounced in the *bc*-plane, we expect that also some of the IR-active modes involving O1 oxygen displacements along the *b* axis and O2 oxygen displacements along the *c* axis will exhibit anomalous behavior around the ferromagnetic transition temperature. According to our assignment, the possible candidates are B_{2u}(4) : 308 cm⁻¹, B_{2u}(6) : 380 cm⁻¹, B_{1u}(3) : 324 cm⁻¹, and B_{1u}(5) : 387 cm⁻¹.

VI. ACKNOWLEDGMENTS

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- [41] To extract information about the phonons we fitted the ellipsometry spectra in Figs. 2-4 using a simple Lorentz model for noninteracting harmonic oscillators. However, as one can see from Figs. 2-4, several phonon bands show weak asymmetry, with flatter rise and steeper fall of the imaginary part of $\tilde{\epsilon}(\omega)$. At the same time the $-1/\tilde{\epsilon}(\omega)$ spectra were found to be fitted fairly well with a Lorentz model. The asymmetry of optical phonon modes suggests mode coupling and interaction damping, which can be accounted for by a phenomenological model for coupled oscillators, $\tilde{\epsilon}(\omega) = \epsilon_\infty + \sum_j S_j \frac{\omega_j^2 - i\omega\sigma_j}{\omega_j^2 - \omega^2 - i\omega\gamma_j}$, with the asymmetry parameters σ_j [40]. We estimate that the level of asymmetry of all TO phonons in YTiO₃ is fairly weak and within the experimental uncertainties in our case.
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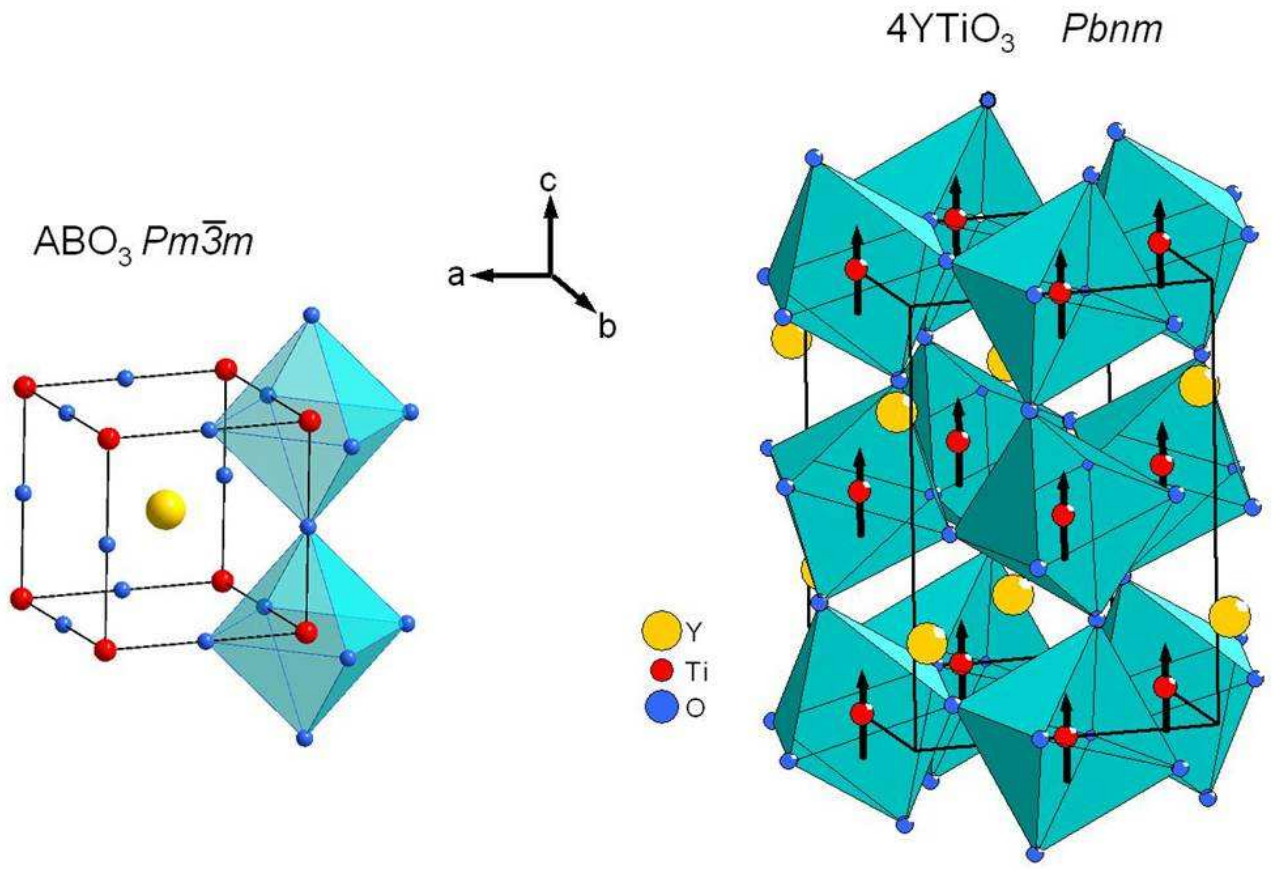


FIG. 1: Crystal structure of cubic perovskite oxides ABO_3 (space group $Pm\bar{3}m$) where A is a rare-earth atom (yellow sphere) and B is a transition-metal atom (red spheres) (left), and related orthorhombic structure of $YTiO_3$ (space group $Pbnm$), with ferromagnetic alignment of Ti spins shown schematically (right).

TABLE I: Wyckoff notations, atomic site symmetries, and irreducible representations for the atoms in orthorhombic $Pbnm$ (D_{2h}^{16}) $YTiO_3$.

Atom	Wyckoff notation	Site symmetry	Irreducible representation
Y	4(c)	C_s^{xy}	$2 A_g + A_u + 2B_{1g} + B_{1u} + B_{2g} + 2B_{2u} + B_{3g} + 2B_{3u}$
Ti	4(b)	C_i	$3 A_u + 3B_{1u} + 3B_{2u} + 3B_{3u}$
O1	4(c)	C_s^{xy}	$2A_g + A_u + 2B_{1g} + B_{1u} + B_{2g} + 2B_{2u} + B_{3g} + 2B_{3u}$
O2	8(d)	C_1	$3A_g + 3A_u + 3B_{1g} + 3B_{1u} + 3B_{2g} + 3B_{2u} + 3B_{3g} + 3B_{3u}$
			$7A_g + 8A_u + 7B_{1g} + 8B_{1u} + 5B_{2g} + 10B_{2u} + 5B_{3g} + 10B_{3u}$
			$\Gamma_{IR} = 7B_{1u} + 9B_{2u} + 9B_{3u}$
			$\Gamma_{Raman} = 7A_g + 7B_{1g} + 5B_{2g} + 5B_{3g}$
			$\Gamma_{silent} = 8A_u$
			$\Gamma_{acoustic} = B_{1u} + B_{2u} + B_{3u}$

TABLE II: Results of the dispersion analysis based on the three-parameter Lorenz model: frequencies and damping parameters ω_{TOj} and γ_{TOj} (in cm^{-1}), and dimensionless oscillator strengths S_j of the TO phonon modes in the the polarized $\tilde{\epsilon}(\omega)$ spectra of the $YTiO_3$ single crystal ($T = 25 \text{ K}$), ϵ_∞ is the core contribution; frequencies and damping parameters ω_{LOj} and γ_{LOj} (in cm^{-1}) of the LO phonon modes in the the $-1/\tilde{\epsilon}(\omega)$ spectra. Definition of ϵ_∞^* is explained in the text.

<i>a</i> axis					<i>b</i> axis					<i>c</i> axis				
ω_{TOj}	γ_{TOj}	S_j	ω_{LOj}	γ_{LOj}	ω_{TOj}	γ_{TOj}	S_j	ω_{LOj}	γ_{LOj}	ω_{TOj}	γ_{TOj}	S_j	ω_{LOj}	γ_{LOj}
154	5.5	0.38	155	4.3	139	5.5	0.53	141	4.0	167	3.8	7.15	196	2.9
209	5.1	3.77	236	5.2	240	8.8	2.36	253	5.0	204	4.9	0.30	216	6.5
316	6.9	2.05	331	3.6	273	7.3	0.60	279	9.8	324	8.6	3.02	346	4.8
335	3.1	0.12	342	8.5	308	5.3	0.51	313	4.6	349	6.2	0.08	354	9.5
344	9.7	0.09	351	6.6	364 ^a	10.0	1.37	369	7.9	387	9.0	2.37	484	7.6
426	7.7	2.78	508	13.2	380	9.9	2.59	438	6.9	545	7.2	0.85	695	7.3
516	10.4	0.07	525	7.5	461	8.8	0.61	496	9.2	595	12.0	0.01	595	12.5
554	7.1	0.61	683	7.1	528	12.5	0.82	689	10.6	—	—	—	—	—
576	11.3	0.02	576	11.2	578	11.2	0.02	577	9.6	—	—	—	—	—
$\epsilon_\infty=3.5$					$\epsilon_\infty=4.0$					$\epsilon_\infty=3.1$				
$\epsilon_\infty^*=4.0$					$\epsilon_\infty^*=4.1$					$\epsilon_\infty^*=3.8$				

^athe splitting of this TO phonon in *b*-axis is probably due to a cross talk of the *a*-axis with strong LO mode at 351 cm^{-1} .

TABLE III: Buckingham potential parameters for the shell-shell and core-shell interactions, the ionic shell charges (Y) (in units of a free electron charge) and force constants (k) in orthorhombic $YTiO_3$, $r_{cutoff} = 19 \text{ \AA}$.

	A(eV)	$\rho(\text{\AA})$	$C(\text{eV} \cdot \text{\AA}^{-6})$
O-O	22764.0	0.1490	20.37
Ti-O1	2902.8	0.2837	00.00
Ti-O2	1240.1	0.3313	00.00
Y-O1	3860.6	0.2944	00.00
Y-O2	12769.1	0.2495	00.00
Ion	Y ($ e $)		k ($\text{eV} \cdot \text{\AA}^{-2}$)
O1 ²⁻	-2.000342		34.4
O2 ²⁻	-3.271589		35.7

TABLE IV: Experimental and calculated properties of YTiO₃. The lattice constants and fractional atomic coordinates present the data for the Czochralski-grown single crystals at T = 293 K [31].

Properties	Experiment	Calculation
$a, \text{\AA}$	5.316(2)	5.3124
$b, \text{\AA}$	5.679(2)	5.6880
$c, \text{\AA}$	7.611(3)	7.6072
$V, \text{\AA}^3$	229.799	229.864
$x/a, y/b, z/c$		
Y	0.9793(1), 0.0729(1), 0.25	0.9778, 0.0692, 0.25
Ti	0.5, 0.0, 0.0	0.5, 0.0, 0.0
O1	0.121(1), 0.458(1), 0.25	0.1185, 0.4485, 0.25
O2	0.691(1), 0.310(1), 0.058(1)	0.6967, 0.3055, 0.0595
Dielectric constants		
ϵ_{11}^0	13.7	13.34
ϵ_{22}^0	13.6	14.02
ϵ_{33}^0	17.7	17.52
ϵ_{11}^∞	3.5-4.0	3.84
ϵ_{22}^∞	4.0-4.1	4.31
ϵ_{33}^∞	3.1-3.8	3.21

TABLE V: Calculated eigenfrequencies $\omega_{\text{TO}j}^{\text{SMC}}$ and components of eigenvectors χ_{ij} of the IR normal modes in orthorhombic YTiO₃. Signs of the components are referred to the atoms with fractional coordinates in the unit cell: Y(0.9793, 0.0729, 0.25), Ti(0.5, 0.0, 0), O1(0.121, 0.458, 0.25), and O2(0.691, 0.310, 0.058). The eigenvector components used in the inversion of Eq. (6) are underlined.

$\omega_{\text{TO}j}^{\text{SMC}}$	Y			Ti			O1			O2		
	x	y	z	x	y	z	x	y	z	x	y	z
$B_{2u}(1):149$	-0.378	0.042	0.000	-0.265	-0.052	-0.005	0.087	-0.005	0.000	0.043	-0.001	-0.104
$B_{2u}(2):203$	0.015	<u>-0.221</u>	0.000	-0.085	<u>0.373</u>	0.094	0.005	<u>-0.052</u>	0.000	0.141	<u>-0.036</u>	0.002
$B_{2u}(3):302$	0.007	-0.178	0.000	0.043	<u>0.057</u>	-0.157	-0.157	<u>0.030</u>	0.000	-0.101	<u>0.144</u>	-0.224
$B_{2u}(4):340$	0.130	<u>0.059</u>	0.000	-0.099	<u>0.011</u>	0.100	0.189	<u>-0.365</u>	0.000	-0.083	<u>0.104</u>	-0.051
$B_{2u}(5):408$	0.215	-0.084	0.000	-0.143	<u>-0.104</u>	0.040	0.262	<u>0.267</u>	0.000	0.063	<u>0.055</u>	-0.071
$B_{2u}(6):446$	0.117	0.115	0.000	-0.041	0.011	-0.381	0.035	-0.114	0.000	0.142	-0.088	-0.054
$B_{2u}(7):497$	-0.069	-0.148	0.000	-0.092	0.011	-0.222	0.072	0.002	0.000	-0.042	0.163	0.227
$B_{2u}(8):614$	-0.076	<u>-0.017</u>	0.000	0.218	<u>-0.129</u>	0.068	0.001	<u>-0.078</u>	0.000	0.235	<u>0.171</u>	-0.015
$B_{2u}(9):546$	0.140	0.040	0.000	-0.288	-0.109	0.076	-0.327	<u>-0.026</u>	0.000	0.072	0.061	0.038
$B_{3u}(1):129$	-0.049	-0.310	0.000	0.114	0.308	0.044	0.005	-0.073	0.000	-0.043	0.065	0.111
$B_{3u}(2):232$	<u>-0.204</u>	0.196	0.000	<u>0.305</u>	0.065	0.177	<u>-0.059</u>	-0.013	0.000	<u>0.005</u>	0.088	-0.105
$B_{3u}(3):291$	<u>-0.223</u>	0.109	0.000	<u>0.040</u>	-0.012	-0.075	<u>0.119</u>	-0.055	0.000	<u>0.169</u>	-0.091	0.212
$B_{3u}(4):372$	-0.141	-0.116	0.000	<u>0.047</u>	-0.113	-0.239	<u>0.289</u>	-0.061	0.000	-0.019	0.119	-0.118
$B_{3u}(5):396$	0.074	0.183	0.000	-0.158	0.174	0.099	0.188	-0.324	0.000	-0.044	-0.005	-0.032
$B_{3u}(6):486$	-0.089	-0.147	0.000	-0.085	0.117	0.039	-0.038	-0.019	0.000	0.197	-0.139	-0.200
$B_{3u}(7):495$	-0.042	-0.067	0.000	-0.100	-0.059	0.330	0.267	0.220	0.000	0.002	0.020	0.022
$B_{3u}(8):652$	<u>0.028</u>	0.100	0.000	<u>-0.180</u>	0.142	-0.067	<u>-0.062</u>	0.119	0.000	<u>0.153</u>	0.242	0.013
$B_{3u}(9):561$	0.004	0.143	0.000	<u>0.029</u>	0.266	-0.171	<u>0.123</u>	0.265	0.000	-0.091	-0.111	-0.036
$B_{1u}(1):151$	0.000	0.000	<u>-0.283</u>	-0.094	0.052	<u>0.167</u>	0.000	0.000	<u>0.131</u>	0.132	-0.155	<u>0.123</u>
$B_{1u}(2):154$	0.000	0.000	-0.012	0.064	-0.442	-0.072	0.000	0.000	0.015	-0.066	-0.116	0.070
$B_{1u}(3):322$	0.000	0.000	0.114	0.125	0.135	<u>-0.275</u>	0.000	0.000	<u>-0.150</u>	0.101	-0.102	<u>0.179</u>
$B_{1u}(4):391$	0.000	0.000	<u>-0.153</u>	0.385	0.092	<u>0.102</u>	0.000	0.000	<u>0.030</u>	-0.148	0.040	<u>0.077</u>
$B_{1u}(5):453$	0.000	0.000	0.108	0.186	0.030	0.124	0.000	0.000	-0.065	0.067	-0.214	-0.203
$B_{1u}(6):549$	0.000	0.000	<u>-0.028</u>	-0.115	-0.005	<u>0.202</u>	0.000	0.000	<u>-0.415</u>	-0.082	-0.024	<u>0.066</u>
$B_{1u}(7):599$	0.000	0.000	-0.029	0.160	-0.157	0.099	0.000	0.000	-0.108	0.244	0.170	0.003

TABLE VI: Suggested assignments of the TO phonon modes in the polarized ellipsometry spectra, $\omega_{\text{TO}j}$, with the normal modes resulting from the shell-model lattice-dynamics calculations, $\omega_{\text{TO}j}^{\text{SMC}}$. Squares of net dipole moments calculated with the eigenvector components χ_{ij} , the formal ionic charges Q_i and the effective dynamical charges Q_i^* are compared with the corresponding values estimated from the experimental phonon parameters $\omega_{\text{TO}j}$ and S_j .

$\omega_{\text{TO}j}(\omega_{\text{TO}j}^{\text{SMC}})$	$[k\omega_{\text{TO}j}^2 S_j]$	$[\sum_i Q_i u_{ij}]^2$	$[\sum_i Q_i^* u_{ij}]^2$	$\omega_{\text{TO}j}(\omega_{\text{TO}j}^{\text{SMC}})$	$[k\omega_{\text{TO}j}^2 S_j]$	$[\sum_i Q_i u_{ij}]^2$	$[\sum_i Q_i^* u_{ij}]^2$
$B_{3u}(1):154$ (129)	0.04	0.09	0.20 ± 0.02	$B_{2u}(1):139$ (149)	0.05	0.001	0.001
$B_{3u}(2):209$ (232)	0.78	0.13	0.78 ± 0.06	$B_{2u}(2):240$ (203)	0.64	0.38	0.64 ± 0.05
$B_{3u}(3):316$ (291)	0.97	1.27	0.97 ± 0.27	$B_{2u}(3):273$ (302)	0.21	0.59	1.04 ± 0.20
$B_{3u}(4):335$ (372)	0.06	0.36	0.40 ± 0.17	$B_{2u}(4):308$ (340)	0.23	0.17	0.23 ± 0.04
$B_{3u}(5):344$ (396)	0.05	0.14	0.53 ± 0.10	$B_{2u}(5):364$ (408)	0.86	1.08	1.78 ± 0.30
$B_{3u}(6):426$ (486)	2.38	0.95	0.91 ± 0.17	$B_{2u}(6):380$ (446)	1.76	0.56	0.95 ± 0.16
$B_{3u}(7):516$ (495)	0.09	0.59	1.03 ± 0.27	$B_{2u}(7):461$ (497)	0.61	0.68	1.22 ± 0.20
$B_{3u}(8):554$ (652)	0.88	0.59	0.88 ± 0.12	$B_{2u}(8):528$ (614)	1.08	0.60	1.08 ± 0.17
$B_{3u}(9):576$ (561)	0.03	0.03	0.01 ± 0.003	$B_{2u}(9):578$ (546)	0.03	0.11	0.20 ± 0.03
$B_{1u}(1):167$ (151)	0.94	0.68	0.94 ± 0.15				
$B_{1u}(2):204$ (154)	0.06	0.20	0.32 ± 0.05				
$B_{1u}(3):324$ (322)	1.50	0.56	1.50 ± 0.20				
$B_{1u}(4):349$ (391)	0.05	0.15	0.21 ± 0.03				
$B_{1u}(5):387$ (453)	1.67	1.68	3.19 ± 0.50				
$B_{1u}(6):545$ (541)	1.19	0.78	1.19 ± 0.20				
$B_{1u}(7):595$ (599)	0.02	0.12	0.21 ± 0.03				

TABLE VII: The dynamical effective charges of the atoms (in units of a free electron charge) calculated from experimental phonon parameters, $\omega_{\text{TO}j}$ and S_j , using the eigenvector components χ_{ij} listed in Table V. Uncertainties of the effective charges are estimated by varying the experimental oscillator strength by $\pm 10\%$.

	Q_Y^*	Q_{Ti}^*	Q_{O1}^*	Q_{O2}^*
a -axis: $B_{3u}(2)\{209\} + B_{3u}(8)\{554\}$ + $B_{3u}(3)\{316\}$	1.5 ± 0.3	5.0 ± 0.2	-2.6 ± 0.4	-1.9 ± 0.2
b -axis: $B_{2u}(2)\{240\} + B_{2u}(8)\{528\}$ + $B_{2u}(4)\{308\}$	3.9 ± 0.4	3.9 ± 0.2	-2.5 ± 0.2	-2.7 ± 0.2
c -axis: $B_{1u}(1)\{167\} + B_{1u}(6)\{545\}$ + $B_{1u}(3)\{324\}$	3.4 ± 0.2	4.6 ± 0.3	-2.3 ± 0.2	-2.8 ± 0.2

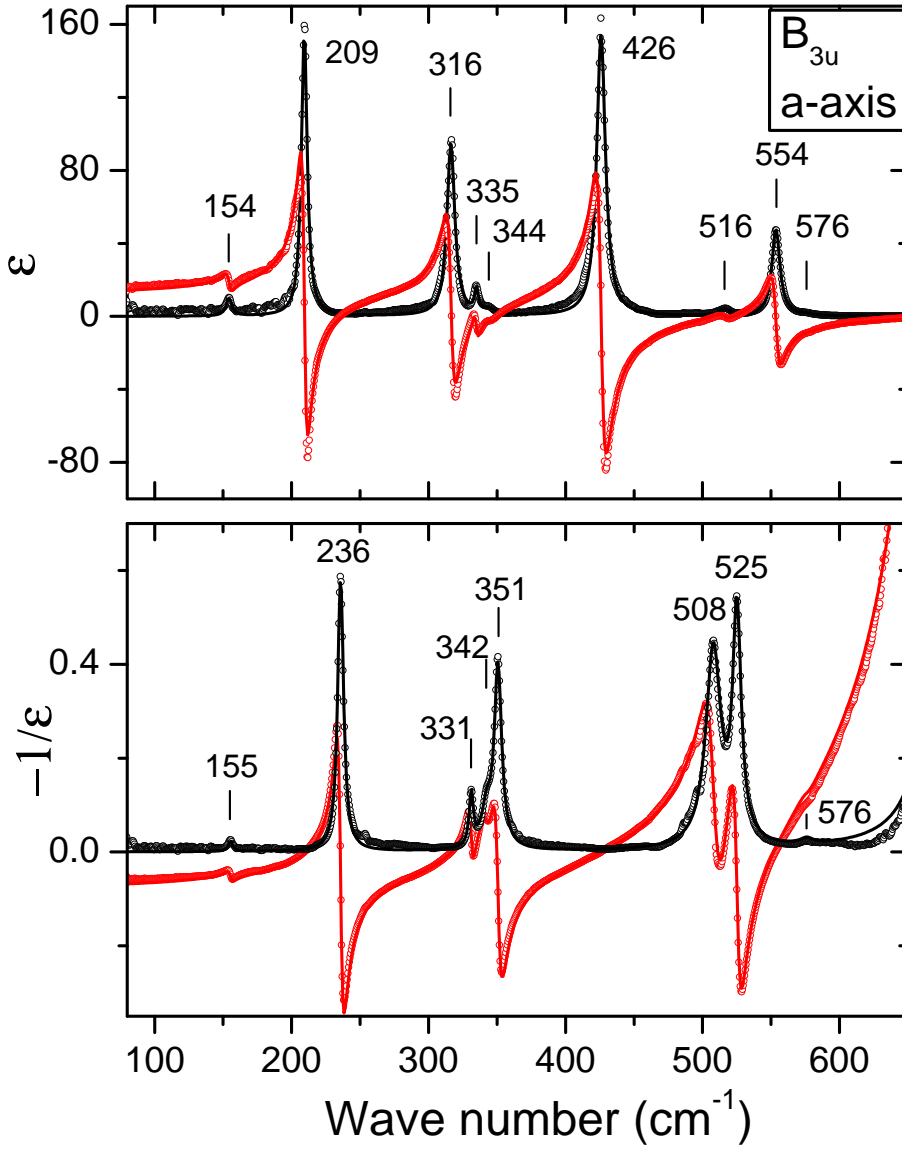


FIG. 2: Real (red open circles) and imaginary (black open circles) parts for the measured a -axis complex dielectric response $\tilde{\epsilon}(\omega)$ and its inverse $-1/\tilde{\epsilon}(\omega)$. The solid curves present the result of the fitting with a set of Lorentzian oscillators, which resonant frequencies $\omega_{\text{TO}j}$ and $\omega_{\text{LO}j}$ are indicated (see also Table II).

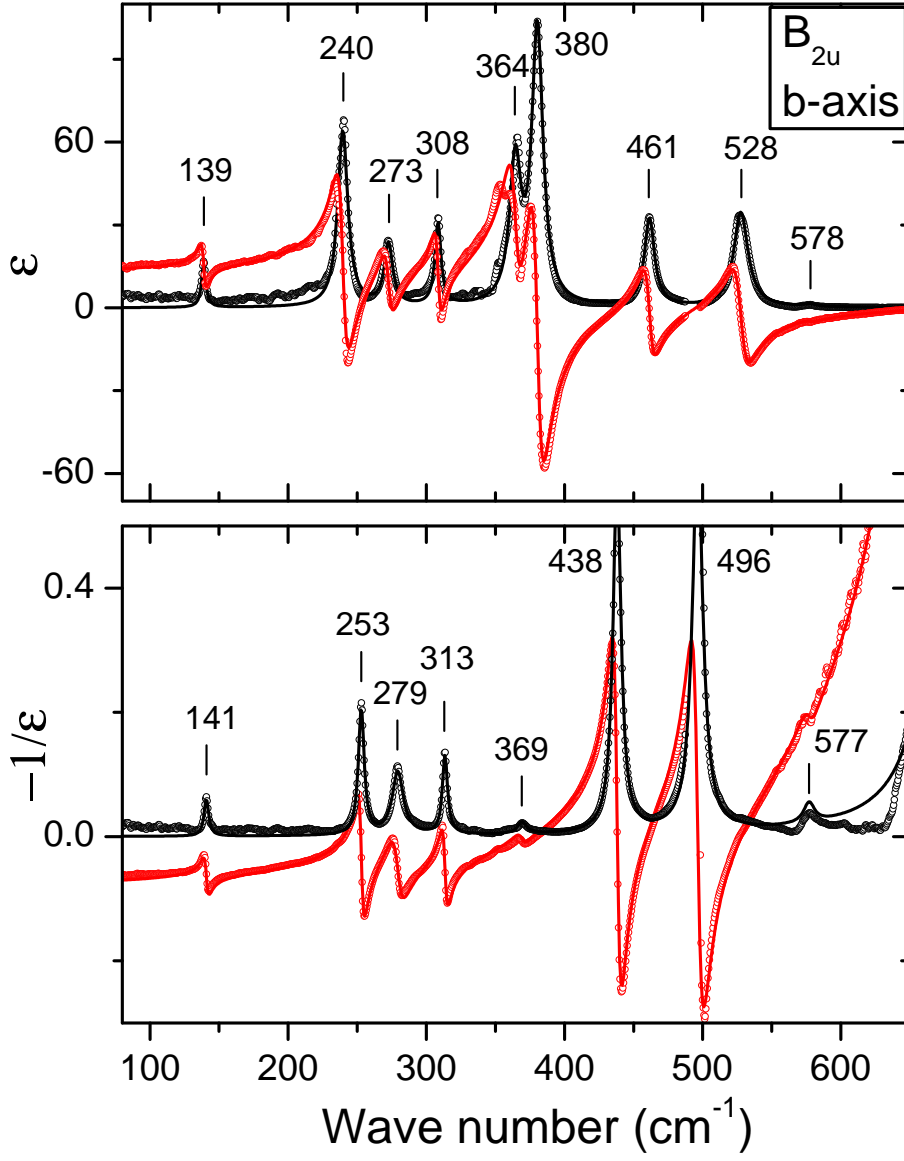


FIG. 3: The same as in Fig. 2 for the b -axis complex dielectric response $\tilde{\epsilon}(\omega)$ and its inverse $-1/\tilde{\epsilon}(\omega)$.

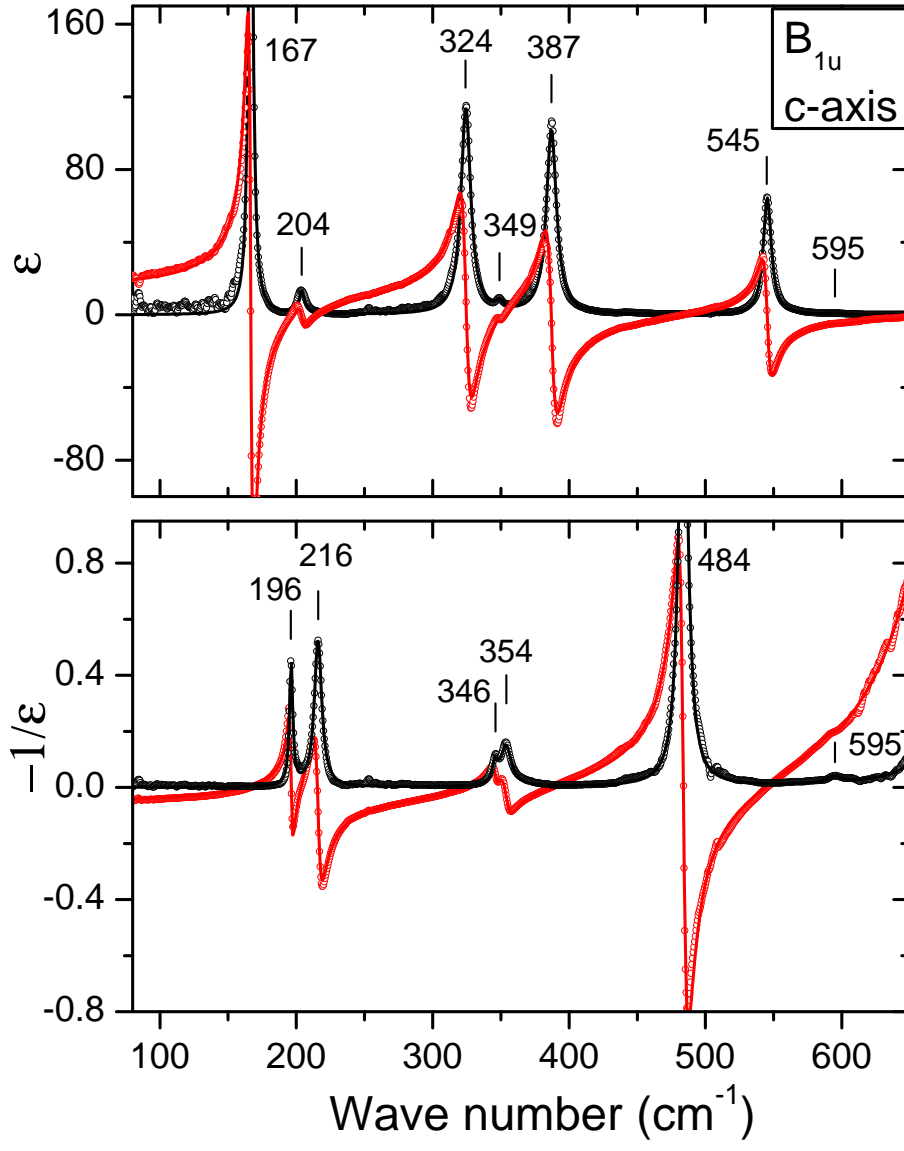


FIG. 4: The same as in Fig. 2 for the *c*-axis complex dielectric response $\tilde{\epsilon}(\omega)$ and its inverse $-1/\tilde{\epsilon}(\omega)$.

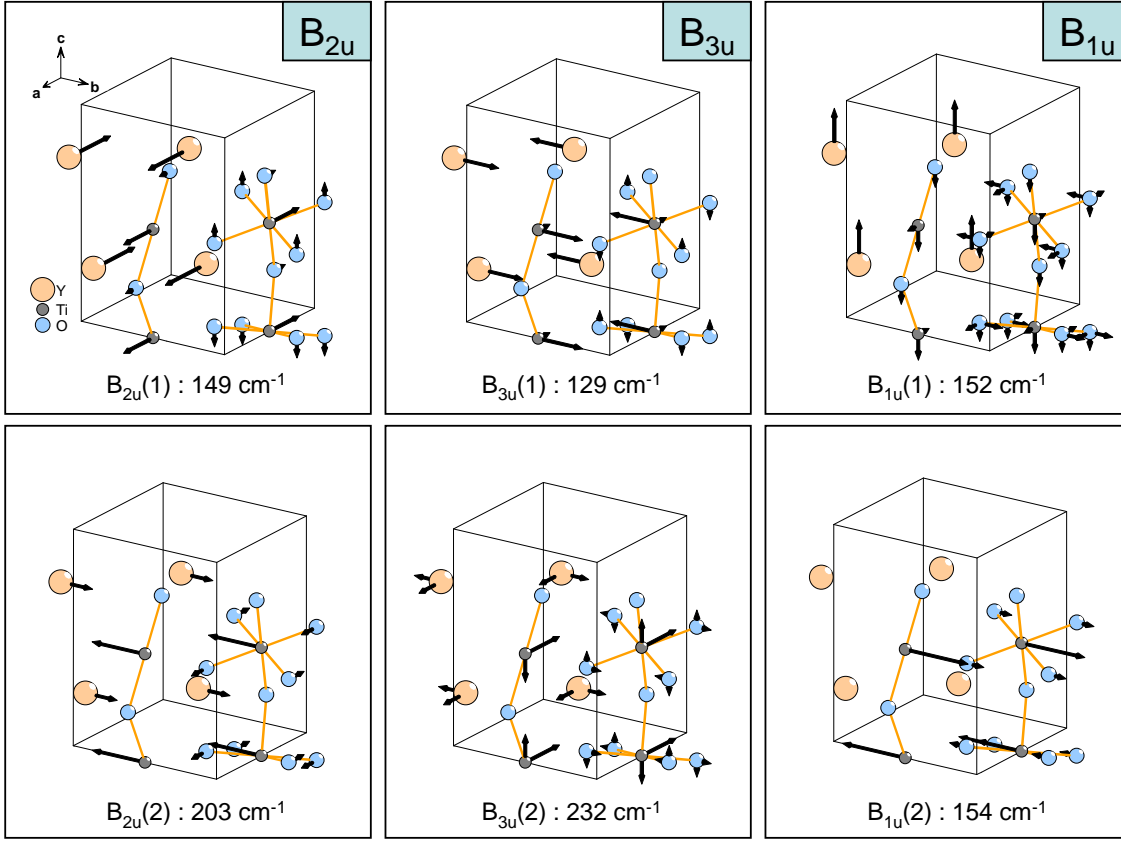


FIG. 5: Eigenfrequencies and eigenvector components of the normal modes of B_{1u} , B_{2u} , and B_{3u} symmetries in orthorhombic YTiO_3 at low frequencies, near the characteristic frequency of the perovskite *external* mode.

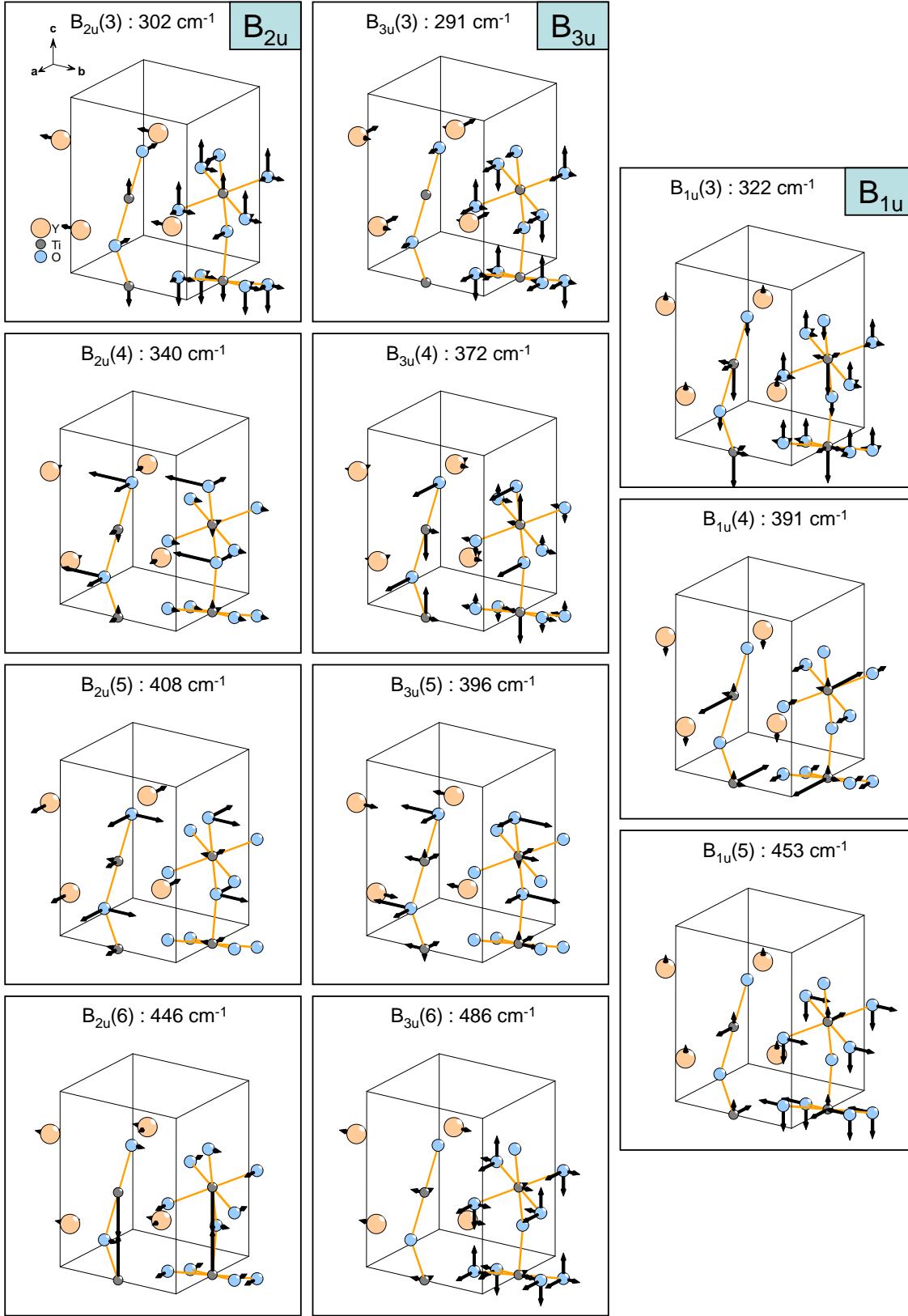


FIG. 6: Eigenfrequencies and eigenvector components of the normal modes of B_{1u} , B_{2u} , and B_{3u} symmetries in orthorhombic $YTiO_3$ near the characteristic frequency of the perovskite *bending* mode.

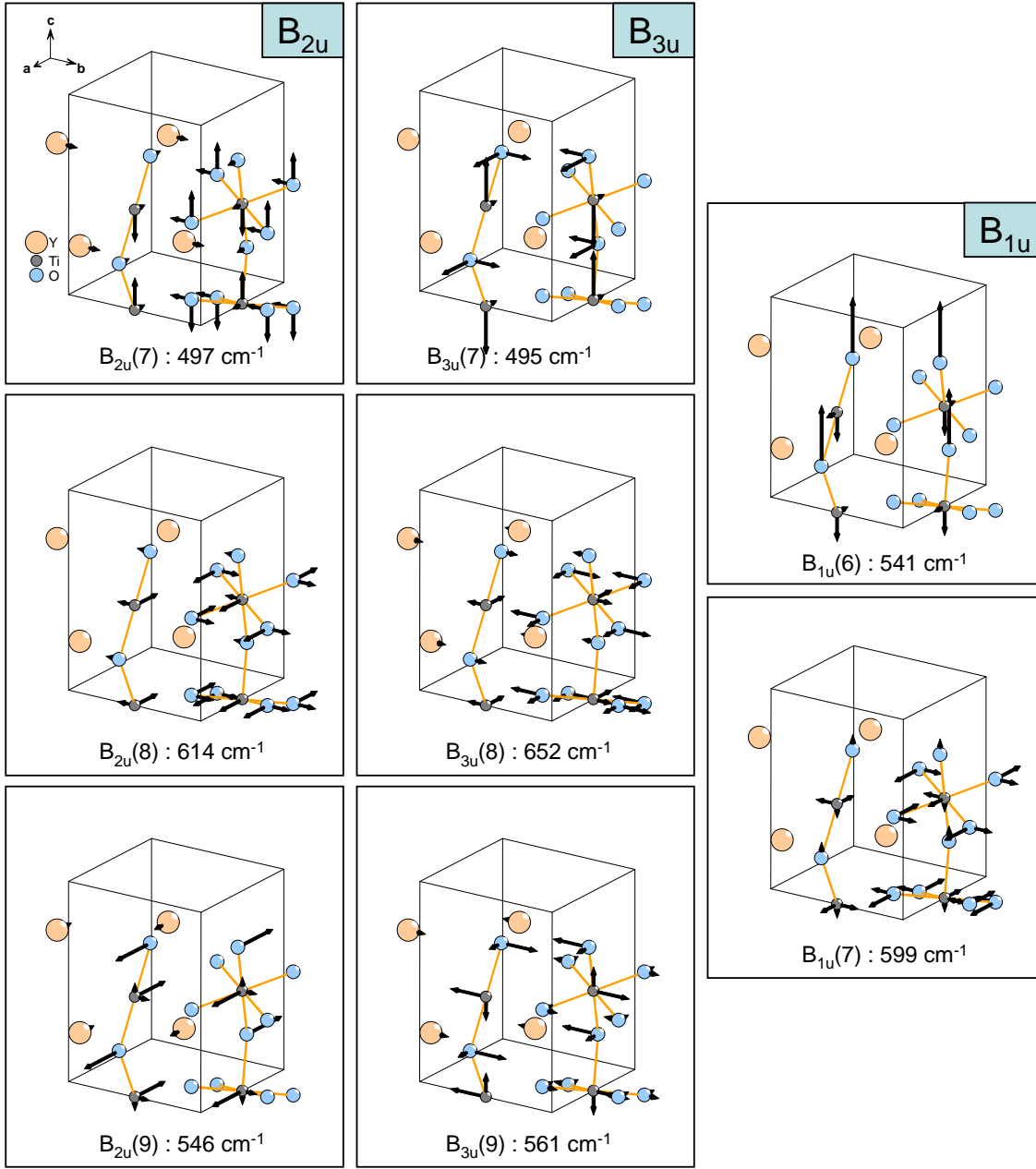


FIG. 7: Eigenfrequencies and eigenvector components of the normal modes of B_{1u} , B_{2u} , and B_{3u} symmetries in orthorhombic YTiO_3 at high frequencies, near the characteristic frequency of the perovskite *stretching* mode.